A pH-, salt- and solvent-responsive carboxymethylcellulose-*g***-poly(sodium acrylate)/medical stone superabsorbent composite with enhanced swelling and responsive properties**

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Abstract. Free-radical graft copolymerization among sodium carboxymethylcellulose (CMC), partially neutralized acrylic acid (NaA), medical stone (MS) and crosslinker *N*,*N*!-methylene-*bis*-acrylamide (MBA) was performed to prepare new carboxymethylcellulose-g-poly(sodium acrylate)/medical stone (CMC-*g*-PNaA/MS) superabsorbent composites. Fourier transform infrared (FTIR) spectra, thermogravimetry- differential scanning calorimetry (TG-DSC) and field emission scanning electromicrsocopic (FESEM) analysis confirmed that NaA had been grafted onto CMC backbone and MS participated in polymerization, and the thermal stability and surface morphologies were improved by the addition of MS. Energy dispersive spectrometer (EDS) and elemental map (EM) analyses revealed the better distribution of MS in the CMC-*g*-PNaA matrix. The incorporation of 20 wt% MS clearly enhanced the water absorption by 100% (from 317 to 634 g/g). The developed composites showed enhanced swelling rate and On-Off switching swelling characteristics in various pH solutions, saline solutions and hydrophilic organic solvents, which represented interesting and reversible pH-, saline- and hydrophilic organic solvent-responsive characteristics. In addition, the composite exhibited intriguing time-dependent kinetic swelling properties in various heavy metal solutions.

Keywords: polymer composites, carboxymethylcellulose, medical stone, superabsorbent, smart-responsive swelling

1. Introduction

Organic/inorganic composite materials have unwaveringly been the subject of great interests because they frequently exhibited desired hybrid performance superior to their individual components and cost-efficient characteristics [1–4]. Among them, the composites derived from natural polysaccharides and inorganic clay minerals are especially focused by virtue of their unique commercial and environmental advantages [5], and such materials have also been honored as the material families 'in greening the $21st$ century materials world' [6].

Superabsorbents are particular 3-D network hydrophilic functional polymer materials that can absorb and conserve considerable amounts of aqueous liquids even under some pressure. Owing to the advantages over conventional absorbents (e.g. sponge, cotton, wood pulp and colloidal silica, etc.), superabsorbents have found potential application in many fields such as agriculture [7, 8], hygiene articles [9], wastewater treatment [10–13], carrier of catalyst [14] and drug delivery carrier [15, 16], etc. However, most of the practically available superabsorbents are petroleum-based synthetic polymer

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with severe environmental impact and high production cost [17], and so the design and development of eco-friendly superabsorbents by introducing natural raw materials have long been desired.

Thus far, many polysaccharides including starch [18, 19], cellulose [20, 21], chitosan [22, 23], alginate [24], and guar gum [25] etc. have been introduced into superabsorbent for improving the performance and environmentally friendly properties. Compared with other natural polymers, cellulose is the most potential one because it is the most abundant natural polymer with renewable, odourless, non-toxic, biodegradable and biocompatible characteristics [26]. However, cellulose is difficult to be directly modified to produce other useful materials due to its poor solubility and reactivity. Carboxymethylcellulose (CMC) is an anionic carboxymethyl ether of cellulose produced by the alkalicatalyzed reaction of cellulose with chloroacetic acid, which has been widely used as a thickening agent and stabilizing agent in industrial fields. The introduced polar carboxyl groups render the cellulose soluble, chemically reactive and strongly hydrophilic, whereas the connatural odourless, tasteless, nontoxic and biodegradable characteristics of cellulose can be retained. So, CMC is advantageous to fabricate eco-friendly polymer materials by its grafting modification [27], and its application in superabsorbent fields becomes attractive and promising. Medical stone (MS) is a special clay mineral that is different from other clays in both structure and properties. MS is an igneous rock composed of silicic acid, aluminum oxide and above 50 kinds of constant and trace elements, and so it was known as 'health-stone' [28]. The main chemical composition of MS is aluminum metasilicate including KAl $Si₃O₈$, NaAl $Si₃O₈$, CaAl₂ $Si₂O₈$, $MgAl₂Si₂O₈$ and FeAl₂Si₂O₈, etc. In such a structure, silica $(SiO₂)$ presents regular tetrahedron with a $[SiO_4]$ configuration, and has a three-dimensional stereo-structure in which aluminum coordinates through oxo-bridging. MS has excellent porosity, multicomponent characteristic, biological activity and safety, and so it shows excellent surface activities, and has been extensively applied in food science, medicine, daily chemical industry, environmental sanitation and wastewater treatment, etc. [29]. However, few information is available on the application of MS as an inorganic additive for the manufacture of eco-friendly superabsorbent composite materials. So, the composite of CMC, NaA and MS was expected to derive superabsorbent materials with improved network structure and swelling properties.

Based on the above background, in the current work, a series of carboxymethylcellulose-*g*-poly (sodium acrylate)/medical stone (CMC-*g*-PNaA/ MS) superabsorbent composites were prepared by the combination of CMC and MS. The structure, thermal stability and morphologies of the developed composites were characterized by Fourier transform infrared spectroscopy (FTIR), thermogravimetric and differential scanning calorimetry analysis (TG-DSC), and field emission scanning electron microscopy (FESEM), energy dispersive spectrometer (EDS) and elemental map (EM) analyses. In addition, the swelling properties, and the stimuli-responsive behaviors of the composite in various medium were evaluated systematically.

2. Experimental

2.1. Materials

Sodium carboxymethylcellulose (CMC, chemically pure, 300~800 mPa·s (25°C)) was from Sinopharm Chemical Reagent Co., Ltd, China. Acrylic acid (AA, chemically pure, Shanghai Shanpu Chemical Factory, Shanghai, China) was distilled under reduced pressure before use. Medical stone (MS) micro-powder (Chinese M-Stone Development Co., Ltd, NaiMan, Inner Mongolia, China) was milled and passed through a 320-mesh screen $($ <46 μ m) prior to use, and the main chemical composition is SiO₂, 68.89%; Al₂O₃, 14.06%; Fe₂O₃, 3.61%; K₂O₂ 3.18%; Na2O, 4.86%; CaO, 1.33%; MgO, 2.59%. The Cation Exchange Capacity (CEC) is 9.91 meq/100 g, the BET specific surface area is $3.915 \text{ m}^2/\text{g}$ and the average pore size is 11.943 nm. Ammonium persulfate (APS, analytical grade, Xi!an Chemical Reagent Factory, China) and *N*,*N*methylene-*bis*-acrylamide (MBA, Chemically Pure, Shanghai Chemical Reagent Corp., China) was used as received. All other reagents used were of analytical grade and all solutions were prepared with distilled water.

2.2. Preparation of CMC-*g***-PNaA/MS superabsorbent composites**

CMC (1.04 g) was dissolved in 30 ml of distilled water at 60ºC in a 250 ml four-necked flask equipped with a mechanical stirrer, a thermometer, a reflux condenser and a nitrogen line to obtain a transparent solution. Afterward, 5 ml of the aqueous solution of the initiator APS (72 mg) was added dropwise to the reaction flask under continuous stirring, and kept at 60ºC for 10 min to generate radicals. 7.2 g of AA was pre-neutralized using 8.0 ml of 8.5 M NaOH solution, and then crosslinker MBA (21.6 mg) and calculated amount of MS powder (0, 0.44, 0.93, 2.08, 3.57 and 5.57 g, respectively) were added under magnetic stirring to form a uniform dispersion. After cooling the reactant to 50ºC, the dispersion was added to the reaction flask, and the temperature was gradually raised to 70°C and kept for 3 h to complete polymerization. Continuous purging of nitrogen was used throughout the reaction period. After being fully washed with distilled water, the obtained gel products were dried to a constant mass at 70ºC, ground and passed through a 40–80 mesh sieve $(180 - 380 \,\mu m)$.

2.3. Measurements of equilibrium water absorption and swelling kinetics

Dry sample (about 0.05 g) with the size of $180~\sim$ 380 µm was adequately contacted with excessive aqueous solution at room temperature for 4 h, until a swelling equilibrium was reached. The swollen samples were separated by a 100-mesh screen and then drained on a sieve for 10 min to remove the excess water. After weighing the swollen samples, the equilibrium water absorption of the superabsorbent was calculated using Equation (1) :

$$
Q_{\text{eq}} = \frac{w_{\text{s}} - w_{\text{d}}}{w_{\text{d}}}
$$
 (1)

where Q_{eq} [g/g] is the equilibrium water absorption calculated as grams of water per gram of sample, which are averages of three measurements; w_d and *w*^s are the weights of the dry sample and waterswollen sample, respectively.

Kinetic swelling behaviors of superabsorbents in each aqueous solution were measured by the following procedure: 0.05 g samples were immersed in 200 ml of aqueous solution for a set period of time. Then, the swollen gels were filtered using a sieve, and the water absorption (Q_t) of superabsorbents at a given time (*t*) can be measured by weighing the swollen and dry samples, and calculated according to Equation (1). In all cases three parallel samples were used and the averages are reported in this paper.

2.4. Evaluation of pH-responsivity

The buffer solutions with various pH values were prepared by combining KH_2PO_4 , K_2HPO_4 , H_3PO_4 and NaOH solution properly and the pH values were determined by a pH meter (DELTA-320). Ionic strengths of all the buffer solutions were controlled to 0.1 M using NaCl. The equilibrium water absorption (*Q*eq) of the composite in various pH buffer solutions was measured by a method similar to that in distilled water. The pH-reversibility of the superabsorbent composites was investigated in terms of their swelling and deswelling between pH 2.0 and 7.2 buffer solution of phosphate. Typically, the sample particle $(0.05 \text{ g}, 180 \text{ m})$ was placed in a 100 mesh sieve and adequately contacted with pH 2.0 buffer solution, 154 mmol/l LiCl, NaCl or KCl solutions and distilled water until reaching equilibrium. Then, the swollen samples were soaked in pH 7.2 buffer solutions, distilled water, and methanol or ethanol for set time intervals. Finally, the swollen samples were filtered, weighed and then calculated the water absorption at a given moment according to the mass change of samples before and after swelling. The consecutive time interval is 15 min for each cycle, and the same procedure was repeated for four cycles. After every measurement, each solution was renewed.

2.5. Determination of network parameter *Mc* The swelling properties of the superabsorbent depend on the crosslinking density, which can be reflected by the average molar mass among the crosslinks, M_c . M_c value is reversibly proportional to crosslinking density and can be calculated by a previously developed method [30, 31]. Typically, *Mc* can be expressed as Equation (2) according to Flory-Huggins theory [32]:

$$
M_c = Q^{5/3} \cdot \frac{D_2 V_1}{0.5 - X_1} \tag{2}
$$

where *Q* represents the equilibrium water absorption of the superabsorbent; D_2 denotes the density of superabsorbent; *V*¹ denotes the molar volume of the solvent used for swelling studies and X_1 is the Flory-Huggins interaction parameter between solvent and superabsorbent. Because it is difficult to determine X_1 by experiment, the linear relationship between X_1 and C (the volume fractions of methanol in methanol/water mixture) was established to determine X_1 value (Equation (3)):

$$
X_1 = K_1 C + K_2 \tag{3}
$$

After simplified above equations, it was concluded that the relations among each parameter follows Equation (4):

$$
D_2 V_1 Q^{5/3} \text{methanol-water} = M_c (0.5 - K_1 C) \tag{4}
$$

The plots of $D_2V_1Q^{5/3}$ methanol-water against *C* give a set of straight lines, and the M_c can be calculated according to the slope of the resulting lines for each superabsorbent.

2.6. Characterizations

FTIR spectra were recorded on a Nicolet NEXUS FTIR spectrometer in $4000-400$ cm⁻¹ region using KBr pellets. The composite sample after swelling in various heavy metal solutions was fully washed and dried before FTIR determination. The surface morphologies, energy dispersive spectrometer (EDS) and elemental map (EM) of the samples were examined using a JSM-6701F Field Emission Scanning Electron Microscope (JEOL) after coating the sample with gold film. Thermogravimetric (TG) and Differential scanning calorimetry (DSC) analyses of samples was studied on a Perkin-Elmer TGA-7 thermogravimetric analyzer (Perkin-Elmer Cetus Instruments, Norwalk, CT), with a temperature range of 25–700°C at a heating rate of 10 °C/min using dry nitrogen purge at a flow rate of 50 ml/min. The samples were dried at 100°C for 4 h to remove the absorbed water before determining TG-DSC curves.

3. Results and discussion 3.1. FTIR spectra analysis

As shown in Figure 1, the characteristic absorption bands of CMC at 1061, 1118 and 1162 cm^{-1} (stretching vibration of C–OH groups) were obviously weakened after reaction and shifted to 1056, 1111 and 1169 cm–1 for CMC-*g*-PNaA. The new bands at 1714 cm^{-1} (C=O stretching of $-$ COOH groups), 1568 cm–1 (asymmetrical stretching vibration of $-COO^-$ groups), 1454 and 1409 cm⁻¹ (symmetrical stretching vibration of –COO– groups) appeared in the spectrum of CMC-*g*-PNaA (Figure 1b). This result indicates that NaA monomers were grafted onto CMC backbone. Compared with the spectrum of CMC-*g*-PNaA, the –C=O stretching of –COOH groups at 1714 cm^{-1} shifted to 1715 cm^{-1} , and the asymmetrical stretching of –COO– groups at 1568 cm⁻¹ shifted to 1572 cm⁻¹ after forming composite. The stretching vibration of (Si)O–H at

Figure 1. FTIR spectra of (a) CMC, (b) CMC-*g*-PNaA, (c) CMC-*g*-PNaA/MS (20 wt%), and (d) MS

Figure 2. Schematic representation of the CMC-*g*-PNaA/ MS superabsorbent composite

 3621 cm⁻¹ can almost not be observed in the spectrum of CMC-*g*-PNaA/MS (Figure 1c, d). The Si–O stretching vibration of MS at 1032 cm^{-1} shifted to 1043 cm⁻¹ after forming composite, but its intensity was obviously weakened. The Si–O bending vibration at 468 cm^{-1} can be observed in the spectrum of CMC-*g*-PNaA/MS with weakened intensity. This information gives direct evidence that MS participated in the graft copolymerization reaction through its active silanol groups [33–35]. Moreover, a schematic structure illustration of the CMC-*g*-PNaA/ MS superabsorbent composite is presented in Figure 2, the MS existed in the polymer matrix can be classified as two kinds: chemically bonding and physically filling [35].

3.2. TG-DSC analyses

Figure 3 showed the TG-DSC curves of CMC-*g*-PNaA hydrogel and CMC-*g*-PNaA/MS (20 wt%) composite. It is obvious that introduction of MS exhibited great influence on the thermal behaviors of the superabsorbents. The weight loss about 6.4 wt% below 251ºC for CMC-*g*-PNaA (corresponding to an endothermic step at 248ºC) and about 4.9 wt% below 254.7ºC for CMC-*g*-PNaA/ MS (corresponding to an endothermic step at 253ºC) can be ascribed to the removal of water absorbed, the dehydration of saccharide rings in the chain of CMC [36]. The successive weight loss about 20.4 wt% (251~432ºC) for CMC-*g*-PNaA and about 17.3 wt% (255~436ºC) for CMC-*g*-PNaA/MS can be attributed to the breaking of C–O–C bonds in CMC chains (corresponding to the

Figure 3. TG-DSC curves of CMC-*g*-PNaA and CMC-*g*-PNaA/MS (20 wt%)

endothermic peaks at 399ºC for CMC-*g*-PNaA and 401ºC for CMC-*g*-PNaA/MS) and the elimination of water molecule derived from the two neighboring carboxylic groups of the grafted PNaA chains due to the formation of anhydride (corresponding to the endothermic peak at 424ºC for CMC-*g*-PNaA and 425ºC for CMC-*g*-PNaA/MS) [37]. However, the endothermic peak of CMC-*g*-PNaA/MS at 425ºC is obviously weaker than that of CMC-*g*-PNaA, and the new endothermic peak at 455ºC also appeared in the DSC curve of CMC-*g*-PNaA/MS. This indicates that the thermal decomposition process in this region was relieved by introducing MS. The weight losses about 20.1 wt% from 432 to 502ºC for CMC-*g*-PNaA (corresponding to a sharp endothermic peak at 482° C) and about 19.1 wt% from 436 to 504ºC for CMC-*g*-PNaA/MS (corresponding to the continuous endothermic peak from 485 to 497ºC) are due to the breakage of main-chain scission and the destruction of crosslinked network structure. Compared with the sharp endothermic peak of CMC-*g*-PNaA, the appearance of continuous endothermic peak in the DSC curve of CMC-*g*-PNaA/MS reveals that the thermal decomposition of the superabsorbent was delayed after incorporating MS. As described above, CMC-*g*-PNaA/MS composite exhibited slower weight-loss and thermal decomposition rate as well as lesser total weight loss than CMC-*g*-PNaA, which indicated the incorporation of MS improved the thermal stability of superabsorbent.

3.3. FESEM, EDS and EM analyses

Figure 4 showed the FESEM micrographs of CMC*g*-PNaA, MS and CMC-*g*-PNaA/MS (20 and 40 wt%)

Figure 4. FESEM micrographs of (a) CMC-*g*-PNaA, (b) MS, (c) CMC-*g*-PNaA/MS (20 wt%), and (d) CMC-*g*-PNaA/MS (40 wt)

superabsorbent composite. As can be seen, CMC-*g*-PNaA only has a smooth and dense morphology (Figure 4a), and MS shows a correspondingly coarse and sheet surface morphology (Figure 4b). After compounding MS with CMC-*g*-PNaA, the surface roughness of the superabsorbent was obviously increased and some pores and gap can be observed (Figure 4c), and the surface roughness of CMC-*g*-PNaA/MS increased with increasing the content of MS (Figure 4d), which implies that introduction of MS contributes to improve the surface morphology of the resultant composite. In addition, this observation gives a direct revelation that MS was uniformly dispersed and almost embedded within the polymer matrix without the flocculation of MS particles, and formed a homogeneous combination of MS with the matrix network.

Figure 5 shows the EDS spectrograms of CMC-*g*-PNaA, MS and CMC-*g*-PNaA/MS (20 and 40 wt%), corresponding to the surface of samples shown in Figure 4a–4d). As can be seen, only the characteristic peaks of C, O and Na elements can be observed in the EDS spectrogram of CMC-*g*-PNaA. After forming CMC*-g*-PNaA/MS composites, the Si, Al, K, Ca, Fe, Na and Mg elements ascribed to MS (Figure 5b) can also be observed, indicating that MS was distributed in the polymer matrix. For further proving this, the elemental maps (EM) for each sample were examined and are shown in Figure 6. It can be noticed that the characteristic O, Si, Al, K, Ca, Fe, Na and Mg elements of MS clearly appeared in the elemental map of the composites, and these elements showed equal distribution in the CMC-*g*-PNaA matrix. This result is consistent with the

Figure 5. EDS curves of (a) CMC-*g*-PNaA, (b) MS, (c) CMC-*g*-PNaA/MS (20 wt%) and (d) CMC-*g*-PNaA/MS (40 wt%)

FESEM observations and EDS results, and reveals the uniform composite of CMC-*g*-PNaA matrix with MS.

3.4. Effects of MS content on M_c **and water absorption**

As shown in Figure 7, the *Mc* of the superabsorbent composite firstly increased and then decreased with increasing the content of MS. A similar tendency can be observed for the water absorption curves of the superabsorbents. The water absorption sharply increased with increasing the content of MS until a maximum absorption (634 g/g in distilled water and 60 g/g in 0.9 wt% NaCl solution) was achieved at 20 wt% MS, and then decreased with further increasing MS content. In comparison with the MS-free superabsorbent hydrogel, the water absorption of the composite increased by 99% after incorporating 20 wt% of MS. This observation indicates that the water absorption of the superabsorbent is mainly dependent on the crosslinking density. The obvious enhancement of water absorption can be attributed to the following reasons: (1) MS may participate in polymerization reaction and act as additional crosslinker. As a result, the polymer network structure was improved and the network void for holding was regularly formed, and the water absorption capacity of the superabsorbent was improved [35, 38]; (2) MS can ionize and release lots of metal cations in contact with water [29], which increased the concentration of electrolyte in interior gel network and enhanced the osmotic pressure difference between gel network and swelling media, and so the water absorption can be enhanced; (3) ionization of

Figure 6. Element area profiles of (a) CMC-*g*-PNaA, (b) MS, (c) CMC-*g*-PNaA/MS (20 wt%) and (d) CMC-*g*-PNaA/MS (40 wt%) at the magnification of 10 000 \times . The small dots denote the distribution of each element in the polymeric matrix

Figure 7. Effects of MS content on M_c (a) and water absorption (b)

MS may generate [–SiO]– groups [29], which can repulse with the negatively charged graft polymer chains and facilitate to the expansion of gel network, and thus contribute to improve the water absorption. However, the excessive addition of MS induced a decrease of water absorption. Because the excessive MS particles may physically fill in the polymer network, which may plug up the network voids for holding water and decrease the hydrophilicity of the composite. For this reason, the water absorption decreased with the increase of the MS content above 20 wt%. It deserves to be mentioned that the water absorption of the composite is till higher than the hydrogel without MS even the content of MS reached 40 wt%, which is extremely favorable to reduce the production cost.

3.5. Swelling kinetics in distilled water

Kinetic swelling curves of the superabsorbents in distilled water were depicted in Figure 8. It can be noticed that the water absorption sharply increased

with prolonging contact time at initial stage $(\leq 1200 \text{ s})$, and then the increasing trend becomes flatter until swelling equilibrium was reached (about 1800 s). In this section, Schott's pseudo second order kinetics model was used to evaluate the swelling kinetics of the superabsorbents (Equation (5)) [39]:

$$
\frac{t}{Q_{\rm t}} = \frac{1}{K_{\rm is}} + \frac{1}{Q_{\infty}}t\tag{5}
$$

where Q_t is the water absorption at a set swelling time *t* [s]; Q_{∞} is the theoretical equilibrium water absorption $[g/g]$; K_{is} is the initial swelling rate constant $[g/(g \cdot s)]$. As shown in Figure 8b), the plots of t/Q_t against *t* showed straight lines with better linear correlation coefficient, indicating that the kinetic swelling behaviors of the superabsorbents follow the pseudo second order model. The values of K_{is} , Q_{∞} and swelling rate constant K_s (= $K_{\text{is}}/Q_{\text{eq}}^2$, [g/(g·s)]) can be determined by fitting experimental data using Equation (5) and calculated by the slope and intercept of the straight lines (Table 1). According

Figure 8. Swelling kinetic curves of the superabsorbents in distilled water (a) and the plots of *t*/*Q*^t against *t* for each superabsorbent (b)

Samples	Q_{eq} [g/g]	Q_{∞} [g/g]	$K_{\text{is}}\left[\text{g}/(\text{g}\cdot\text{s})\right]$	$\sqrt[k]{K_s} \cdot 10^{-5}$ [g/(g·s)]	
$ CMC-g-PNaA $	318	315	2.2783	2.2530	0.9999
$ CMC-g-PNaA/MS (5 wt%)$	424	429	4.6092	2.5639	0.9999
$ CMC-g-PNaA/MS $ (10 wt%)	549	562	3.8069	1.2631	0.9998
$ CMC-g-PNaA/MS (20 wt%)$	633	648	2.8820	0.7193	0.9995
$ CMC-g-PNaA/MS $ (30 wt%)	514	529	2.8636	1.0839	0.9995

Table 1. Swelling kinetic parameters for CMC-*g*-PNaA/MS composites with various amounts of MS

^{*}Initial swelling rate constant K_{is} ; **Swelling rate constant $K_s = K_{is}/Q_{eq}$ ²

to the resultant data, it can be concluded that the initial swelling rate of the superabsorbents decreased in the order of CMC-g-PNaA/MS (5 wt\%) > CMC*g*-PNaA/MS (10 wt%) > CMC-*g*-PNaA/MS (20 wt%) > CMC-*g*-PNaA/MS (30 wt%) > CMC-*g*-PNaA. This result reveals that incorporation of MS also enhanced the initial water absorption rate of the superabsorbents besides water absorption capability. It can also be noticed that the swelling rate constant (K_s) in the whole swelling process follow the order of CMC-*g*-PNaA/MS (5 wt%) > CMC-*g*-PNaA > CMC-*g*-PNaA/MS (10 wt%) > CMC-*g*-PNaA/MS (30 wt%) > CMC-*g*-PNaA/MS (20 wt%), indicating the moderate introduction of MS (5 wt%) is favorable to improve the swelling rate. However, the change trend of K_s is not consistent with that of K_{is} . This is because that the water absorption (Q_{eq}) of the superabsorbent composites with 10, 20 and 30 wt% of MS is higher than CMC-*g*-PNaA hydrogel. This means that the polymer chains need to move greater distance for reaching the swelling equilibrium, so the time of reaching equilibrium for CMC-*g*-PNaA/MS (10, 20 and 30 wt%) is relatively longer than that of CMC-*g*-PNaA.

3.6. pH-responsive characteristics

It can be noticed from Figure 9 that the water absorption of the composite is almost zero at pH 2.0 for both CMC-*g*-PNaA hydrogel and CMC-*g*-PNaA/ MS composite, but the water absorption rapidly increased with enhancing the pH of external solution, and almost keep constant in the pH range of 6– 12. The water absorption follows the order of CMC-*g*-PNaA/MS (20 wt%) > CMC-*g*-PNaA/MS (10 wt\%) > CMC-g-PNaA/MS (30 wt\%) > CMC*g*-PNaA at pH 7.2. In acidic medium, the –COO– groups on the polymer chain converted to –COOH groups. For one thing, the electrostatic repulsion among negatively charged –COO– groups decreased and the expansion of polymer network was restricted; for another, the increase of number of –COOH

Figure 9. Variation of swelling ratio for (a) CMC-*g*-PNaA, (b) CMC-*g*-PNaA/MS (30 wt%), (c) CMC-*g*-PNaA/MS (10 wt%) and (d) CMC-*g*-PNaA/MS (20 wt%) superabsorbents at the buffer solution with various pHs. The inset is the On-Off switching behavior as reversible pulsatile swelling $(pH = 7.2)$ and deswelling $(pH = 2.0)$ of CMC-*g*-PNaA/MS (20 wt%) composite. The time interval between pH changes is 15 min.

groups strengthened the intramolecular and intermolecular hydrogen bonding interaction and enhanced the physical crosslinking degree. Thus, the polymer network swells little at low pH. For evaluating the reversibility of pH-responsivity, the swellingdeswelling cycles were evaluated between the buffer solutions of pH 2.0 (Off) and 7.2 (On). It can be obviously observed that the composite shrunken at pH 2.0 can recover to relatively higher swelling ratio at pH 7.2. After four swelling-deswelling periods between pH 2.0 and 7.2, the intriguing On-Off switchable and reversible swelling effect was observed. This implies that the pH-responsivity of the superabsorbent composite is highly reversible and the introduction of MS effectively enhanced the responsivity of the superabsorbent.

3.7. Salt-responsive characteristics

Figure 10 depicts the circular swelling-deswelling behaviors of the superabsorbent composite between distilled water and 154 mmol/l of LiCl, NaCl and

Figure 10. The reversible On-Off switching swelling behaviors of the composite between distilled and 154 mmol/l of LiCl, NaCl and KCl solutions

KCl solutions. As can be seen, the water-swollen superabsorbent composite can shrink and loss the absorbed water when it was contacted with salt solutions, but the shrunken gel can rapidly recovered in distilled water. The switching salt stimuliresponsive characteristics were observed. As well known, the osmotic pressure difference between internal gel network and external swelling media acts as the driving force for the swelling of superabsorbent; it decreases with adding polyelectrolyte to the solution. Also, the ultimate screen effects of cations on negative charges of polymer chains increased with increasing polyelectrolyte in the solution. These factors induced the deswelling of the composite. After four circles of swelling (On) deswelling (Off), the superabsorbent still keeps excellent responsive properties, indicating the swelling of the superabsorbent is switched and reversible.

3.8. Hydrophilic organic solvents-responsive characteristics

The phase transitions of hydrogels have aroused attention since the collapse of polyacrylamide hydrogels in aqueous solution of acetone was observed [40]. The unique phase transition behavior of superabsorbent to organic solvents endows it with stimuli-responsive characteristics. Figure 11 represents the swelling (On)-deswelling (Off) cyclic curves of the superabsorbent composite between distilled water (On) and methanol or ethanol (Off). It was noticed that the composite exhibited higher water absorption in distilled water, but the absorbed water was sharply lost when it was fully contacted with

Figure 11. The reversible On-Off swelling behaviors of the composite between distilled water and methanol or ethanol

hydrophilic organic solvent. The great shrinkage of the water absorption in hydrophilic organic solvents can be attributed to the decrease of the solubility parameter and dielectric constant as well as the interaction between the polar groups of organic solvent and the ionic groups in the polymer [41]. The solubility parameter of water is $23.4 \text{ (cal/cm}^3)^{1/2}$, but the value of methanol and ethanol is only 14.5 and 12.7 $\text{(cal/cm}^3)^{1/2}$. The great decrease of solubility parameter caused the rapid shrinkage of water absorption [42]. Likewise, the dielectric constant of water is 78.54, but the dielectric constant of methanol and ethanol is only 32.63 and 24.3. As described previously [41], the dielectric constant of the solution directly affects the ionization degree of ionic groups, and the larger the dielectric constant of the solution, the bigger the osmotic pressure of the gel. Thus, the sharp decrease of dielectric constant from hydrophilic organic solvents to water certainly causes the reduction of water absorption. In addition, the polarity of water is larger than methanol and ethanol. The lower polarity of ethanol and methanol is responsible for the easier collapse of composite network in contrast to water [43]. These indicate that the collapse of the composite is a result of the total depletion of water due to the competitive extraction of water from the solvated polymeric network. It can also be observed from Figure 11 that the shrunken gel can rapidly recovered when it was immersed in distilled water, and no obvious slack after four On-Off cycles. This indicates that the hydrophilic organic solvent-responsive properties of the composite are switching and reversible.

3.9. Time-dependent characteristics in heavy-metal solutions

Effects of heavy metal saline solutions on kinetic swelling behaviors of the composite were evaluated in NiCl_2 , CuCl₂ and ZnCl₂ solutions at the concentration of 2, 5 and 10 mmol/l (Figure 12). An intriguing time-dependent swelling behavior was observed for the composite in each saline solution. The water absorption of the composite increased with prolonging contact time, reached a maximum absorption (127, 63 and 40 g/g for 2, 5 and 10 mmol/l of Ni^{2+} ; 126, 65 and 39 g/g for 2, 5 and 10 mmol/l of Cu^{2+} ; and 130, 70 and 41 g/g for 2, 5 and 10 mmol/l of Zn^{2+} , respectively) with further prolonging contact time and then decreased until the swelling almost disappeared. Also, the required time of reaching the maximum absorption was reduced with increasing the concentration of saline solution (12, 8 and 5 min for 2, 5 and 10 mmol/l of each saline solution, respectively). The time-dependent swelling effect can be attributed to the following reasons depicted in Figure 13. The superabsorbent composite contains numerous –COOH and –COO– functional groups, which can complex with heavy metal cations to form an additional ionic crosslinking [12, 44]. So the effective crosslinking degree of composite network increased with prolonging the contact time. Also, the complex of –COO– with heavy metal cations would decrease electrostatic repulsion among negatively charged polymer chains and reduce the expansion degree of the hydrogel network. As a result, the absorbed water would be expelled out from the network void and the swollen hydrogel tends to shrink. After reaching the maximum swelling, the swollen composite network gradually collapsed and the initially absorbed water was squeezed out of the network under this action. For proving the complex action, the FTIR spectra of the composite after swelling in 5 mmol/l $NiCl₂$, CuCl₂ and $ZnCl₂$ solutions for 0.5, 5. 10. 30 and 120 min were determined and are shown in Figure 14. As can be seen, the characteristic absorption bands of $-COO^-$ groups at 1572 cm⁻¹ (Figure 1c) shifted to 1571 (Ni^{2+} , 0.5 min), 1568 (Ni^{2+} , 5 min), 1561 (Ni²⁺, 10 min), 1558 (Ni²⁺, 30 min) and 1556 cm⁻¹ (Ni²⁺, 120 min); to 1570 (Cu²⁺, 0.5 min), 1569 (Cu²⁺, 5 min), 1563 (Cu²⁺, 10 min), 1560 (Cu²⁺, 30 min) and 1557 cm^{-1} (Cu²⁺, 120 min); to 1571

Figure 12. Kinetic swelling curves of CMC-*g*-PNaA/MS (20 wt%) in 2, 5 and 10 mmol/l of (a) $NiCl₂$, (b) $CuCl₂$ and (c) $ZnCl₂$ solutions. The insets are the enlargement of the curves ranged from 0– 60 min.

 $(Zn^{2+}, 5 \text{ min})$, 1570 $(Zn^{2+}, 10 \text{ min})$, 1568 $(Zn^{2+}, 5 \text{ min})$ 30 min) and 1563 cm⁻¹ (Zn^{2+} , 120 min) after swelling in heavy metal saline solutions for different interval, respectively. The shift of absorption band to low wavenumber indicates that Ni^{2+} , Cu^{2+} and Zn^{2+} ions were complexed with –COO– groups [44] and the additional crosslinking in hydrophilic network was formed, and so the swollen gel network tends to deswell.

Figure 13. Proposed mechanism for the swelling–deswelling of the composite in heavy metal solutions

Figure 14. FTIR spectra of the superabsorbent composite after swelling in 5 mmol/l NiCl₂, CuCl₂ and ZnCl₂ solutions for (a) 0.5 min, (b) 5 min, (c) 10 min, (d) 30 min, (e) 120 min

4. Conclusions

For developing new kinds of superabsorbent composites with improved structure, properties and environmentally friendly characteristics, reducing the excessive consumption of petroleum resource and minimizing the pollution from the industrial polymers, a series of cellulose-based superabsorbent composites containing MS were prepared by free-radical graft copolymerization technique. FTIR analysis revealed that the NaA monomers had been grafted onto the macromolecular chains of CMC, and MS participated in the polymer reaction and combined with the network by chemically crosslinking and physically filling models. MS led

to a better dispersion in the CMC-*g*-PNaA matrix as shown by EDS and EM analyses. The incorporation of MS and the formation of composite structure clearly improved the surface morphologies, thermal stability, swelling capabilities, and swelling rate of the developed composite. The superabsorbent composite exhibited excellent switching swellingdeswelling behaviors between pH 2.0 and 7.2 buffer solution, between distilled water and 154 mmol/l LiCl, NaCl and KCl solutions, between distilled water and hydrophilic organic solvents, and the obvious pH-responsive, saline-responsive and hydrophilic organic solvent-responsive properties were presented. In addition, the intriguing timedependent swelling behaviors were observed in various heavy metal saline solutions, which are caused by the strong complexation action between hydrophilic –COO– groups and heavy metal ions. As described above, the superabsorbent composite based on renewable, low-cost and biodegradable natural CMC and abundant MS exhibited improved swelling properties, excellent pH-responsive, saltreversible, hydrophilic organic solvent-reversible properties and intriguing time-dependent swelling behavior, which can be used as potential candidate for water-manageable materials or drug delivery system.

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